

Influence of preparation conditions on superconducting properties of Bi-2223 thin films

N T MUA^{1,2}, A SUNDARESAN^{3,*}, N K MAN² and D D DUNG²

¹Police Department of Fire Prevention Fire Fighting and Rescue, No. 2, DinhLe Road, Hoan Kiem District, Hanoi, Vietnam

²International Training Institute for Materials Science (ITIMS), Hanoi University of Science and Technology, No. 1, Dai Co Viet Road, Hanoi, Vietnam

³Chemistry and Physics of Materials Unit and International Centre for Materials Science, Jawaharlal Nehru Centre for Advanced Scientific Research, Bangalore 560 064, India

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Abstract. We report electrical transport properties of $\text{Bi}_2\text{Sr}_2\text{Ca}_2\text{Cu}_3\text{O}_{10+x}$ (Bi-2223) superconducting thin films fabricated by pulsed-laser deposition on SrTiO_3 substrate. The aim of the study was to investigate the influence of preparation conditions such as deposition temperature (T_S), annealing time (t_A) and deposition rate (r). A critical temperature (T_c) as high as 110 K and critical current density (J_c) of 6.2×10^6 A/cm² at 20 K were obtained for $T_S = 760^\circ\text{C}$, $t_A = 4$ h and $r = 1.5$ Å/s. We also investigated the effect of Li doping on Bi-2223 thin films. Li intercalation results in high resistive onset transition temperature and the resistivity shows broadening in magnetic field that increases with field. The large broadening of resistivity curve in magnetic field suggests that this phenomenon is directly related to the intrinsic superconducting properties of the copper oxide superconductors. The sudden drop in J_c at relatively low magnetic field ($H < 0.5$ tesla) is due to the effect of Josephson weak-links at the grain boundaries.

Keywords. Bi-2223; thin film; pulsed laser deposition; critical current density.

1. Introduction

From both scientific and technology points of view, it is important to study how the superconducting transition temperature changes with thickness of film, substrate and also to examine critical thickness for the occurrence of superconducting states. In the case of high-temperature superconductors (HTS), the dimensionality has also been an interesting subject for understanding the mechanism of superconducting phenomenon. One of the important materials that has been investigated extensively is $\text{Bi}_2\text{Sr}_2\text{Ca}_2\text{Cu}_3\text{O}_{10+x}$ (Bi-2223) superconducting thin film (Simon 2003). Several fabrication techniques such as pulsed-laser deposition (PLD) (Dijkkamp *et al* 1987), molecular beam epitaxy (LPE) (Fujino *et al* 2001), co-evaporation (Cui *et al* 2001), metal organic chemical vapour deposition (MOCVD) (Doudkowsky *et al* 1997) have been used to prepare Bi-2223 films. Among these methods, PLD method is known to be suitable for fabricating films with complex stoichiometry.

Thin films of Bi-cuprates fabricated by a pulsed laser deposition (PLD) method show promising superconducting properties. Applying this technique, one could obtain thin

films of single-phase Bi-2212 and Bi-2223. However, it was found that fabrication of Bi-2223 thin films was more difficult than that of Bi-2212. Critical temperatures of 69 and 78 K were obtained for as-deposited Bi-2212 and Bi-2223, respectively at the deposition temperatures of 820 and 850 °C (Nori-hisa *et al* 1999), though a better crystalline film was obtained at a lower temperature of 750 °C for Bi-2212 and Bi-2223. The substrate temperature was also found to influence surface morphology of the film. $\text{Bi}_2\text{Sr}_2\text{Ca}_{n-1}\text{Cu}_n\text{O}_{2(n+2)+\delta}$ ($n = 1, 2, 3$) thin films are prepared in general on vicinal SrTiO_3 substrates by pulsed-laser deposition. Optimization of substrate temperature, laser fluence and post-annealing conditions produces single-phase, well-oriented and smooth films. On (0 0 1) SrTiO_3 substrates, c -axis oriented Bi-2212 films with $J_c = 2 \times 10^6$ A/cm² at 60 K and $T_{c,zero} = 82$ K are obtained (Rössler *et al* 2001). Bi-2223 film grown at a temperature of 725 °C displays a sharp transition of 10 K, which is close to the best result (11 K) obtained by Ishii and Hatano (2000) and 13 K obtained by Matthiesen *et al* (1991).

Further, several studies of the effect of Li intercalation (Kambe *et al* 1996, 1998) on the nominal composition of Bi-2223 have been reported, where it was observed that increase in the quantity of lithium led to the reduction of the volume fraction of Bi-2223 phase. On the other hand,

*Author for correspondence (sundaresan@jncasr.ac.in)

transport measurements have shown that the increase of lithium content leads to the improvement in transport critical current density, J_{ct} (Mihalache *et al* 2001). The systematic studies of LiF (Mihalache *et al* 2003a, b) and LiCl (Minh *et al* 2003) doped Bi-2223 systems revealed that a small quantity of lithium compound helps in the

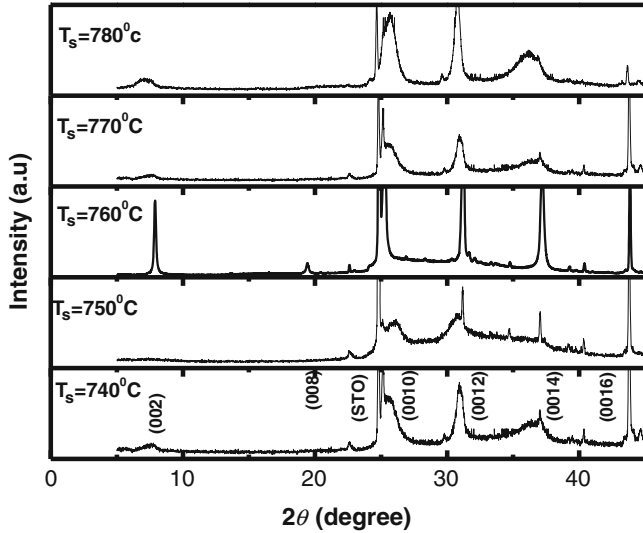


Figure 1. XRD patterns of Bi-2223 thin films on SrTiO₃ substrate deposited at various substrate temperatures: 740, 750, 760, 770 and 780 °C.

formation of high- T_c superconducting phase Bi-2223. The transport and intergrain critical current densities (J_{ct} and J_{cj}) vs the doping level exhibit a maximum and the critical temperature T_c of 110 K superconducting phase Bi-2223 reaches 120 K (Mihalache *et al* 2001, 2003a, b).

In this paper, we report effect of preparation conditions on the properties of Bi₂Sr₂Ca₂Cu₃O_{10+x} thin films prepared by the PLD method using BSCCO target which was prepared by the solid-state reaction method.

2. Experimental

The target was prepared from the starting materials, Bi₂O₃, SrCO₃, CaCO₃ and CuO, by the solid-state reaction method. The stoichiometric amount of these oxide powders were mixed thoroughly and calcined at 840 °C for 24 h in air. This step was repeated at 850 °C for 72 h in air with several intermittent grindings to improve the homogeneity. The calcined powder was compressed into a disk of 15 mm in diameter and thickness of 4 mm with the use of polyvinyl alcohol as a binder. The compacted powder disk was heat-treated at 860 °C for 24 h. Before deposition, (100) SrTiO₃ (STO) substrates were annealed in air at 1000 °C for 1 h. The excimer laser ($\lambda = 248$ nm) was operated at 300 mJ/pulse for 30 min for each deposition. Substrate was mounted on the heater in the cross-area of PLD plume caused by the beam and its temperature (T_s) was in the range of 740–780 °C.

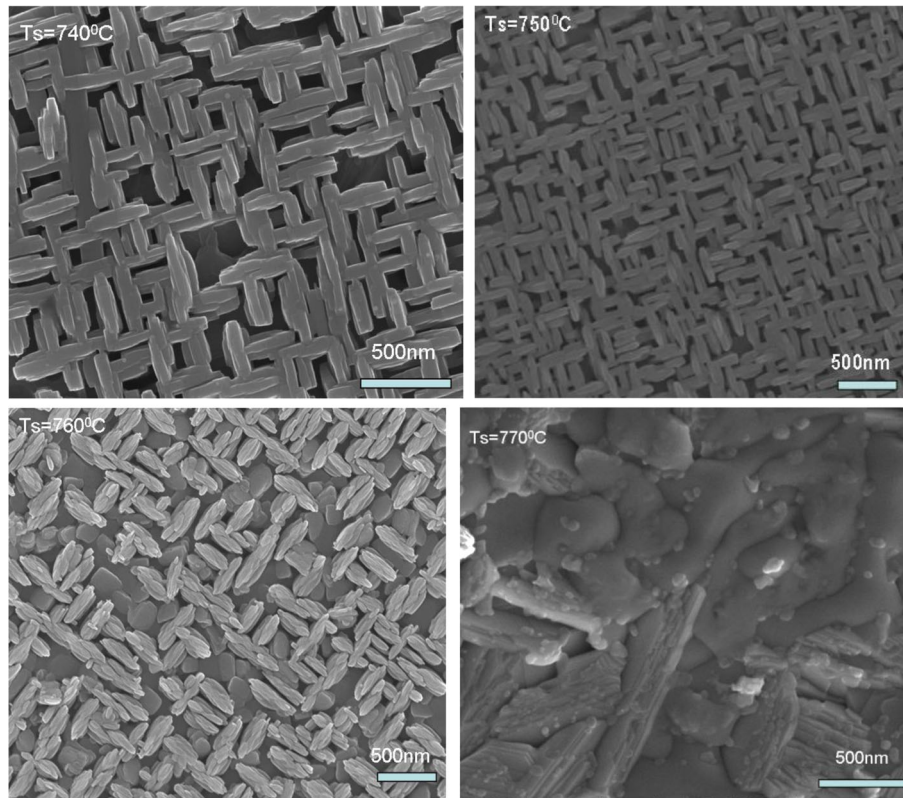


Figure 2. SEM micrograph of Bi-2223 films prepared at various temperatures.

The oxygen pressure was kept at 0.25 mbar. The films were annealed at 680 °C in the range of 1–5 h and cooled down to room temperature in oxygen pressure of 1000 mbar. The film thickness was about of 300 nm. To compare the superconducting properties, we prepared Bi-2223 films with various amounts of Li intercalation. The same preparation conditions were followed for Li intercalated films.

D.C. electrical resistance measurements were carried out using the standard four-probe method with silver alloy solders as electrical contacts. The superconducting transition temperature (T_c) was determined as the midpoint between 90 and 10% of the resistive transition. The structure and surface morphology of the films were studied by X-ray diffraction and field emission scanning electron microscope (FESEM), respectively. Magnetic measurements were carried out using the vibrating sample magnetometer option in the physical property measuring system (PPMS, quantum design). The critical current density J_c of the film was calculated by Bean model

$$J_c = 60a|\Delta M|/b(3a - b),$$

where a and b are the length and width ($a > b$, in cm) of the sample plane perpendicular to the applied magnetic field. ΔM is the difference of the magnetization (emu/cm^3) between the field-up and field-down branches.

3. Results and discussion

3.1 Properties of Bi-2223 thin film on STO substrate

X-ray diffraction (XRD) pattern for Bi-2223 thin films at various substrate temperatures are shown in figure 1. The presence of (002) peak in XRD pattern of thin films corresponds to a well-crystallized orthorhombic phase with c -axis

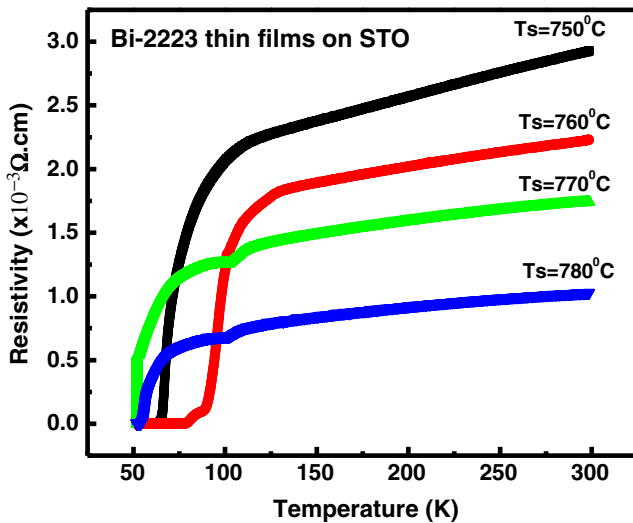


Figure 3. Resistivity of Bi-2223 thin films at different substrate temperatures.

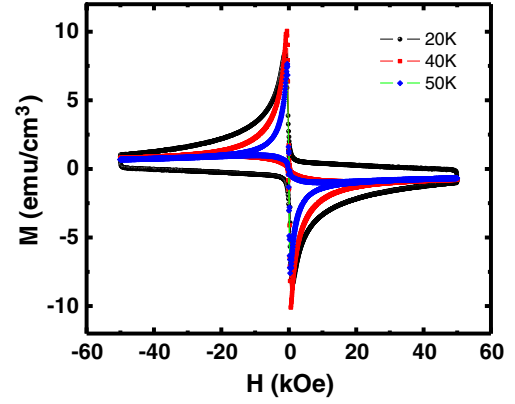


Figure 4. Magnetic field dependence of current density at different temperatures.

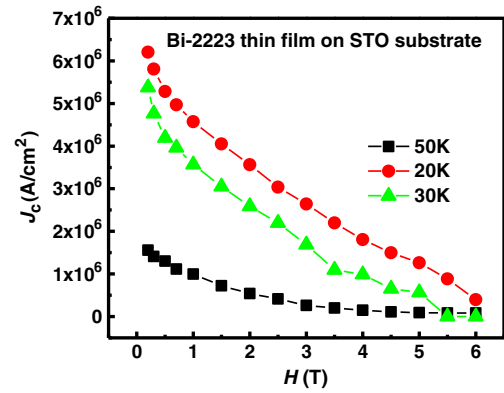


Figure 5. M–H of Bi-2223 thin films at 20, 30 and 50 K.

oriented perpendicular to the film surface. The obtained value for c -axis length is for the film deposited at substrate temperature of 760 °C, 37.68 Å which is comparable to that reported in the literature. This confirms that Bi-2223 thin films grown on single crystalline (100) STO substrate is of high structural quality. This also reflects perfection of orientation of different c -axis oriented blocks of the film which is relatively normal to the substrate. It is observed that there exists a trace amount of CuO as a secondary phase. Images of scanning electron micrographs of Bi-2223 films prepared at different temperatures are shown in figure 2. It can be clearly seen that there are particles with Bi-2223 composition on the surface of all the films. Many grain boundaries and voids could be seen for the films on STO due to large lattice mismatch (+2.31%). Relatively, smooth films with a surface roughness, grain size of around 200 nm and a low particulate density were obtained on SrTiO₃ as seen in figure 2.

Figure 3 shows temperature dependence of resistivity for the films at different substrate temperatures. All the films show a metallic behaviour in the normal state. A typical resistivity curve measured on *ex-situ* annealed films, as shown in

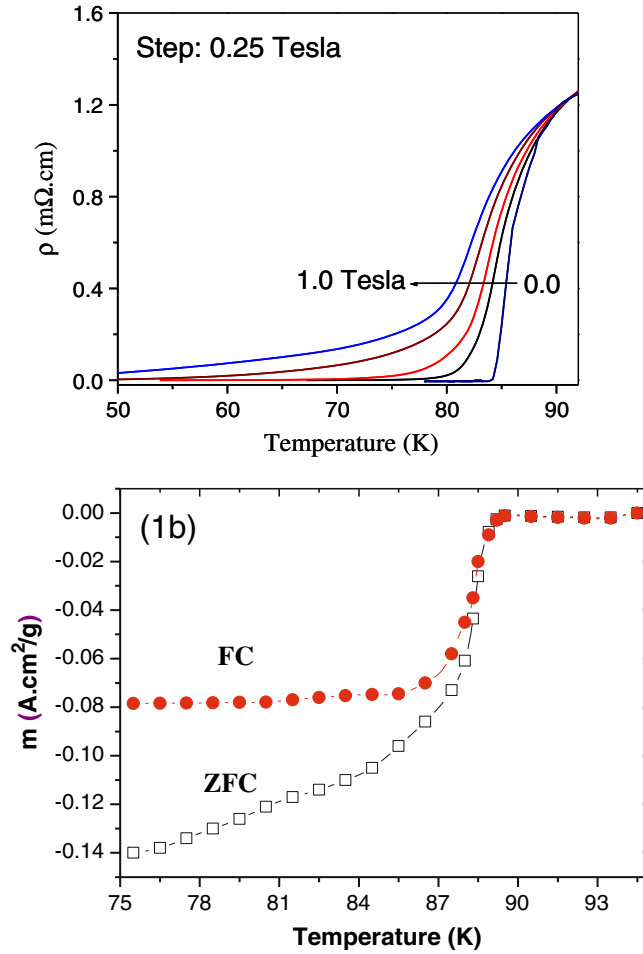


Figure 6. ρ vs T of BSCCO thin films for different magnetic fields and FC and ZFC of BSCCO thin films measurement at $H = 100$ Oe.

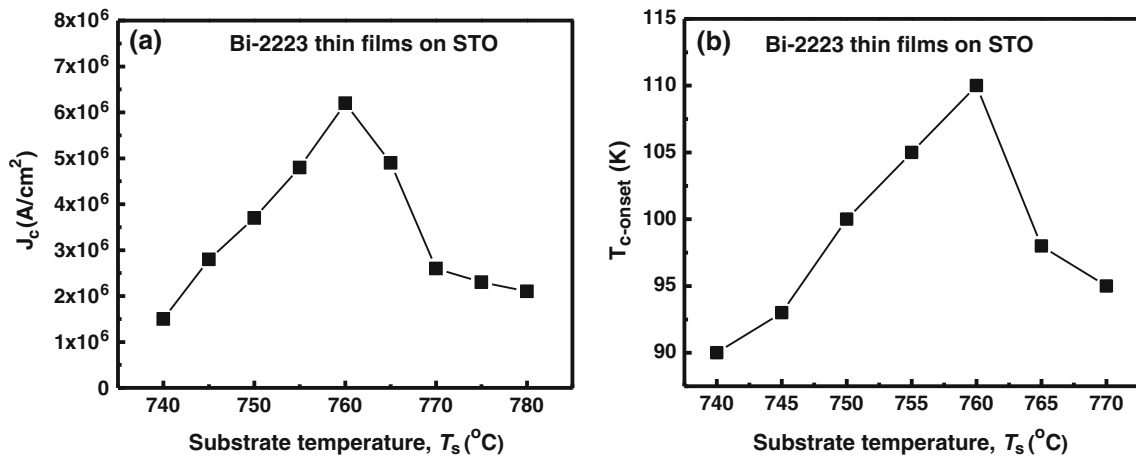


Figure 7. Effect of deposition temperature (T_s) on J_c and T_c in Bi-2223 thin films.

figure 3, displays a broad transition with a transition temperature, $T_{c-onset} = 110$ K and zero resistance at 95 K. The broad transition may be due to the presence of CuO impurity. The resistivity tail is very short compared with the width

of transition. This could be due to the presence of inhomogeneities in the oxidation state of grains of superconducting materials of good structural quality. It has been reported that the critical temperature of Bi-2223 materials is very

much dependent on the oxygen concentration. Since oxygen incorporation depends sensitively on the deposition conditions, slight changes of the conditions during growth could generate variations in oxygen concentration. In particular, we would like to point out that growing superconductor layer induces change in the infrared reflectivity of the substrate film system, which could result in a shift in substrate temperature. It is interesting to notice that the resulting inhomogeneities in oxygen concentration would lead to variations in the structural parameters.

The critical current density at various temperatures and magnetic field obtained from the analysis of the magnetization data (figure 4) using Bean model is shown in figure 5. The value of J_c at 20 K and 0.5 T is 6.2×10^6 A/cm² that decreases with increasing magnetic field and temperature. In this configuration, Lorentz force is perpendicular to the *ab*-plane and vortices induced by the applied magnetic field have to cross the superconducting CuO₂ planes for flux flow to occur. It is shown that J_c decreases rapidly when the field is applied parallel to *c*-axis of the film. The sudden drop in J_c as soon as a magnetic field is applied ($H < 0.5$ T) is due to the effect of Josephson weak links at the grain boundaries (Minh

et al 2003). At higher applied magnetic field ($H > 0.5$ T), J_c decreases slowly due to the flux pinning inside the grains. From the critical current density, J_c , the pinning force density F_p can be calculated by using the equation:

$$F_p = J_c \times H.$$

Resistivity as a function of temperature at different magnetic fields up to 1 Tesla is shown in figure 6(a). The results can be summarized as follows: (i) At zero field, the transition is sharp and has a zero-resistance temperature, T_{c0} of 90 K; (ii) the resistive transition in the presence of the magnetic field exhibits a broadening, which increases with increasing magnetic field. The large broadening of resistivity curve in magnetic field suggests that this phenomenon is directly related to the intrinsic superconducting properties of this kind of oxide superconductors. The resistivity in the normal state, above T_c , shows a strong para-conducting behaviour. The magnetization data of BSCCO thin film obtained under zero-field cooling (ZFC) and field-cooling (FC) processes are shown in figure 6(b). It is shown that the diamagnetic transition temperature (T_c) of BSCCO thin film is 95 K. We calculated H_{c2} from the formula:

$$H_{c2} = \sqrt{2\kappa H_C}, H_{c2} \sim \kappa H_C; \kappa = 0, 8,$$

in which

$$H_C = \left[\frac{2}{\mu_0} (g_n - g_s) \right]^{\frac{1}{2}}.$$

3.2 Effect of deposition temperature (T_S)

Heating of the substrate during deposition was required to provide thermal energy for the deposited atoms to migrate on the surface and to arrange themselves in the energetically favoured lattice. A higher substrate temperature (T_S) can provide activation energy for adatoms to occupy the position of potential minima and to increase surface and volume diffusion enhancing recrystallization due to the coalescence

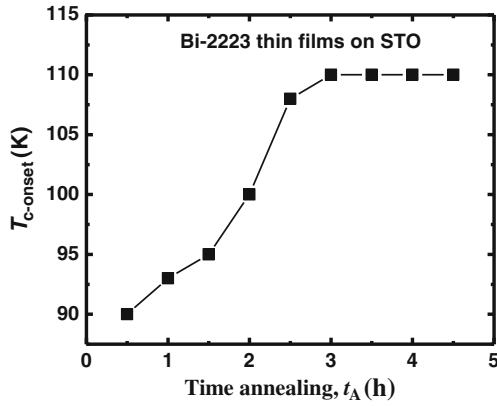


Figure 8. Effect of annealing time (t_A) on T_c in Bi-2223 thin films.

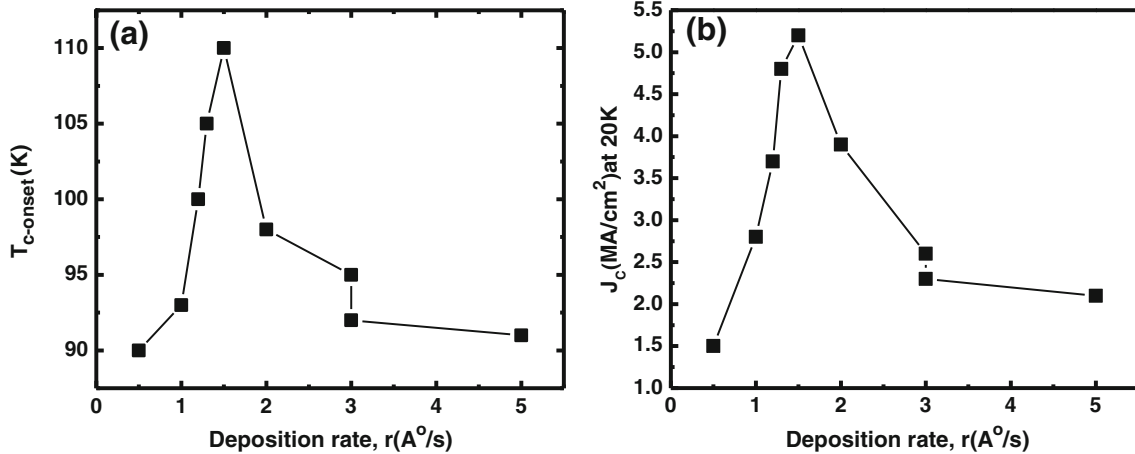


Figure 9. Effect of deposition rate (r) on T_c (a) and J_c at 20 K (b) of Bi-2223 thin films on STO substrate.

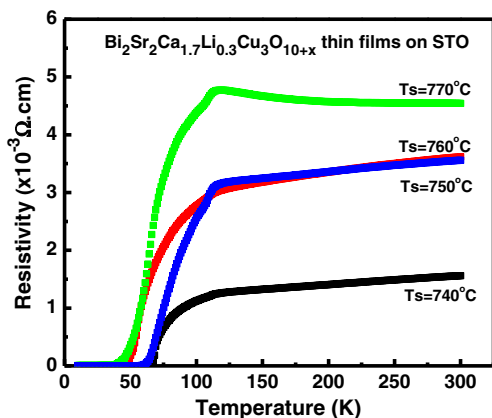


Figure 10. Temperature dependent resistivity of Li-intercalated Bi-2223 thin films prepared on SrTiO₃ substrate at different substrate temperatures.

of islands. On the other hand, higher T_s could cause a serious interdiffusion between film and substrate, which could degrade superconducting properties of the films.

In the present study, effects of T_s on properties of Bi-2223 thin films have been investigated, while the other parameters were kept constant: time of annealing, $t_A = 3$ h and deposition rate, $r = 2 \text{ \AA/s}$. As shown in figure 7, the temperature region between 740 and 780 °C provides an optimum superconducting transition temperature (T_c) and critical current density (J_c). These temperatures are high enough to allow sufficient surface migration and interdiffusion of the atoms for the formation of the desired orthorhombic Bi-2223 crystal structure. It shows that the optimal transition temperature, $T_c = 110$ K, was obtained at T_s of around 760 °C, where the highest critical current density, $J_c = 5.2 \times 10^6 \text{ A/cm}^2$ at 20 K was obtained.

3.3 Effect of annealing time (t_A)

It is well known that the superconductivity of Bi-2223 is strongly affected by oxygen content. The annealing at 680 °C in oxygen is believed to be important for controlling oxygen content of the films. Figure 8 shows transition temperature of Bi-2223 films as a function of annealing time. After 3 h of annealing, the optimal critical temperature of 110 K is reached. Further annealing does not influence the transition temperature and oxygen content, 5 h of reduction treatment, the value is same 3 h.

3.4 Effect of deposition rate (r)

Figure 9 is the dependence of T_c and J_c as a function of Bi-2223 deposition rate ($T_s = 760$ °C, $t_A = 3$ h). These results showed that the optimal deposition rate is 1.5 Å/s. At lower deposition rate, the films' surface was characterized by voids between grains. At higher deposition rate, outgrowths appear which increase in number with further increase of deposition

rate. The outgrowth has deleterious effect on T_c and J_c of Bi-2223 films.

3.5 Effect of Li intercalation in Bi-2223 thin films

Figure 10 depicts typical temperature dependence of electrical resistivity of Li substituted Bi-2223. A similar metallic trend in the normal state was obtained. Li intercalation results in a considerable increase in $T_{c\text{-onset}}$ (115 K). It is also well known that Li-doping lowers melting temperature of this phase, so the decrease in oxygen content would be expected. It can be seen that a linear decrease of resistivity at lower substrate temperatures produces a sharp drop with narrow transition. It should be mentioned that as Li concentration increased, the normal state resistivity of sample decreases and zero-resistivity temperature increases.

4. Conclusions

Bi-2223 thin films of good quality with perfect c -axis orientation have critical temperature of 110 K and critical current density of $6.2 \times 10^6 \text{ A/cm}^2$ at 20 K in 0.5 T. The critical temperature is increased to 115 K for Li intercalated Bi-2223 thin films. The resistive transition in the presence of magnetic field exhibits a broadening induced by the magnetic field and the broadening increases with increasing field. The sudden drop in J_c at a relatively lower magnetic field is applied ($H < 0.5$ T) due to the effect of Josephson weak-links at the grain boundaries. These films have a sufficiently good quality to make the high-performance microwave components. Considering the flux-pinning behaviour, critical current density strongly depends on the applied magnetic field. Large critical current values are obtained in high magnetic fields which are of interest for applications.

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